

## Article

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# **Graphene quantum dots: in the crossroad of graphene, quantum dots and carbogenic nanoparticles**

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## **Abstract**

The scientific and technological importance of graphene quantum dots (GQDs) is directly related to their nanoscopic nature that endows remarkable photo-physical properties and colloidal stability in a variety of solvents. GQDs combine characteristics arising from their graphitic structure, their carbogenic origin and their quantum nature. They are considered as the environmentally benign alternatives of heavy metal based quantum dots, given that not only are they synthesized following green strategies, but they also exhibit minimal toxicity. GQDs are systematically explored in printing, energy harvesting, bioimaging, catalysis, optoelectronics and sensing applications.

## **1. Introduction**

As their name "graphene quantum dots" (GQDs) implies, this emerging class of nanomaterials is placed in the crossroad of the electronically conductive highway of graphene, the photoluminescent (PL) avenue of quantum dots and the green, widely unexplored, pathway of carbogenic nanoparticles. The term "dot" refers to their nanoscopic nature that endows remarkable optical properties<sup>[1-4]</sup> comparable to the best performing fluorescent materials. Due to their advanced PL properties, GQDs are systematically explored in solar cells, optoelectronics, catalysis, cell imaging and sensing.

While pure graphene (an atomic monolayer of carbon perfectly arranged in a honeycomb conformation) is not emissive, energy band gaps can be engineered by introducing structural defects<sup>[5•]</sup> in a graphene plane or via its quantum confinement towards 1D (graphene nanoribbons<sup>[6]</sup>) or OD (GQDs) geometries. The PL behavior of GQDs can be tuned from ultraviolet to near infrared depending on the size, shape, edge effects, functional groups, heteroatom doping and  $sp^2$  carbon fraction.

In addition to their supreme PL behavior, GQDs show a number of attractive characteristics. First, they exhibit minimal toxicity for humans and the environment (see section 5.1), in contrast to their heavy metal based counterparts. Second, they are synthesized in bulk from abundant starting materials by low-cost strategies. Third, because GQDs are dispersible in water and organic solvents, they are readily integrated into standard industrial manufacturing.

The progress achieved so far with respect to the synthesis, characterization and applications of GQDs has been reviewed in a number of recent reports<sup>[1-4]</sup>. Here we attempt to present a compact synopsis of the current state of understanding in GQDs highlighting their graphitic structure, their carbogenic origin and their quantum nature.

Strictly speaking, GQDs refer to nano-fragments of atomically thick graphene, however the term is also used to describe few-layer thick nanodisks of graphite, graphite oxide quantum dots and partially reduced graphite oxide quantum dots. We note that oftentimes in literature, the terms carbogenic nanoparticles, carbon dots, carbon quantum dots and GQDs are used interchangeably<sup>[7]</sup>. Here we refer to carbogenic nanoparticles (C-dots) as a broad class of highly PL materials that includes GQDs, along with other types of carbon-rich particles with partially or predominantly amorphous cores.

## 2. The graphitic structure

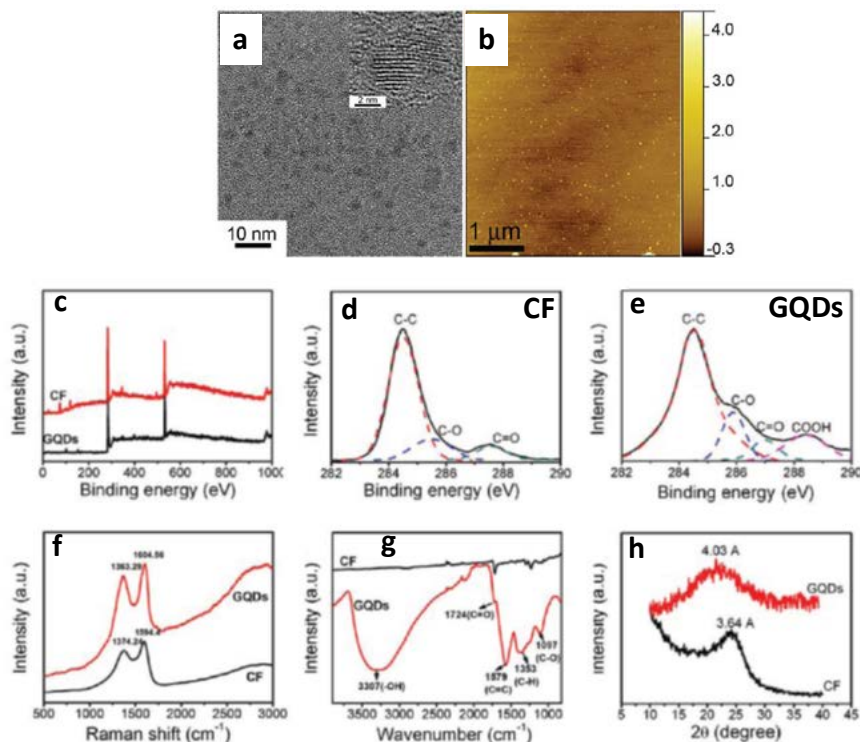
Monodispersed GQDs are prepared by all-organic synthesis starting from polyphenylene dendrimers<sup>[8]</sup>, alkyne containing reactive molecules<sup>[9]</sup> or via self-assembly and carbonization of unsubstituted hexa-peri-hexabenzocoronene precursors<sup>[10]</sup>. Alternatively, well-defined GQDs derived by ring opening of fullerenes that are strongly adsorbed to Ru metal terraces, undergo fragmentation and gasification at elevated temperatures to generate surface-stabilized carbon clusters that coalesce towards uniform structures<sup>[11]</sup>.

A distinct advantage of GQDs is that they can be produced in bulk based on the chemical oxidation of carbon-rich sources such as coal<sup>[12]</sup>, carbon black<sup>[13]</sup>, graphite<sup>[14]</sup>, carbon fibers (CF)<sup>[15]</sup>, CNT<sup>[16]</sup> and fullerenes<sup>[17]</sup>. Alternatively, electrooxidation of CNTs<sup>[18]</sup> and graphite<sup>[19]</sup> leads to the formation of hydroxyl and oxygen radicals (produced by the electrolysis of the solvent) that attack the honeycomb lattice on defect and edge sites, facilitating the release of GQDs.

GQDs with relatively narrow size distribution between 1 to 4 nm and a predominant zigzag edge structure are derived by chemical oxidation and cutting of micro-sized pitch based CF<sup>[15]</sup>. The TEM images (Fig. 1a) reveal a lattice spacing of 0.24 nm, while the Atomic Force Microscopy (AFM) topographic heights (Fig. 1b) indicate that GQDs are composed by 1-3 graphene layers. X-ray Photoelectron Spectroscopy (XPS) patterns of GQDs and CF (Fig. 1c-e) display the C1s peak at 284.8 eV and the O1s peak at 532 eV and suggest a higher oxygen content for GQDs. Analysis of the C1s peak of CF and GQDs (Fig. 1d and 1e, respectively) reveals the enhanced presence of C = C, C–O, C = O bonds and –COOH groups on the surface of GQDs.

The Raman spectra of GQDs and CF (Fig. 1f) are governed by the two characteristic G and D bands of the graphitic and disordered carbon, respectively. The G peak is attributed to the  $E_{2g}$  phonon at the Brillouin center, while the D peak corresponds to the breathing mode of  $sp^2$  centers and becomes active in proximity to an  $sp^3$  defect<sup>[20]</sup>. The relative intensity of the D over G band,  $I_D/I_G=0.9$ , implying a pronounced graphitization degree. Fourier transform infrared (FTIR) spectra (Fig. 1g) provide further evidence for the evolution of –OH, C=O, C=C, C–H and C–O in GQDs. Those groups function as a self-passivating layer that render GQDs dispersible in a variety of solvents. The XRD profile of GQDs (Fig. 1h) shows a broad peak at 4.0 Å compared to a sharp

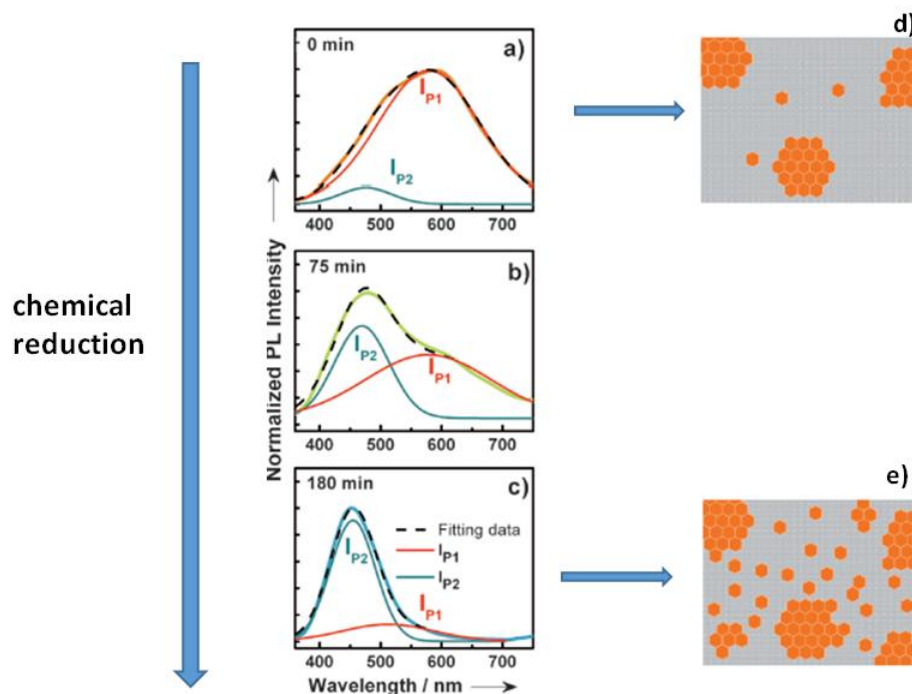
peak at  $3.6\text{\AA}$  for the initial CF. The larger interlayer spacing found for GQDs is consistent with the generation of oxygen-containing groups during oxidation and exfoliation of CF.



**Figure 1.** GQDs released from carbon fibers (CF) via chemical oxidation; (a) TEM image, (b) AFM image, (c,d,e) XPS spectra, (f) Raman spectra, (g) FTIR spectra and (h) XRD patterns. Reprinted with permission from ref. 15. Copyright 2012 American Chemical Society.

### 3. The quantum nature and its limitations

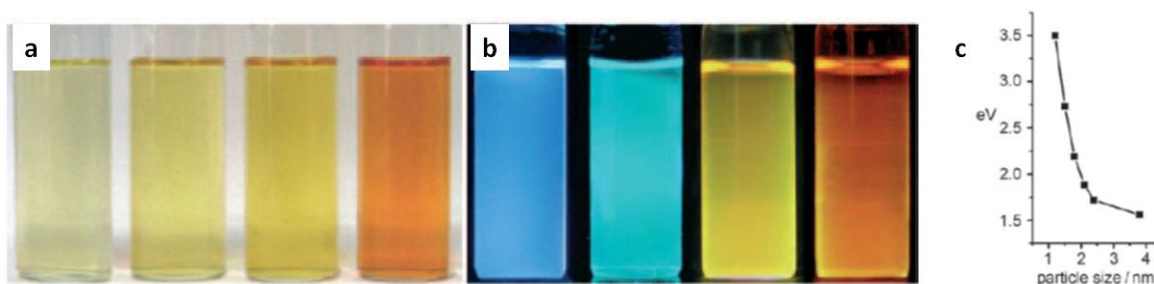
Graphite oxide (GO) consists of a graphene plane bearing numerous oxygen-rich groups, a situation that can be seen as  $sp^2$  carbon islands dispersed within a  $sp^3$  matrix. The PL spectra of aqueous suspensions of GO can be deconvoluted into two Gaussian bands ( $I_{p1}$  and  $I_{p2}$  in Fig. 2 a,b,c), suggesting the parallel action of two distinct emissive contributions<sup>[21••]</sup>. The relative intensities of those contributions significantly change during the photothermal reduction (via xenon lamp irradiation) of GO that results in a dramatic increase of the  $sp^2$  carbon content from 25% of the starting GO (derived via a modified Hummers method) all the way to 69% after 3h exposure.



**Figure 2.** PL spectra of aqueous suspensions of GO at reduction times; (a) 0 min, (b) 75 min and (c) 180 min. Note that each plot can be fitted using two Gaussian curves. (d and e). Graphics showing the birth of small and isolated  $sp^2$  islands during the gradual chemical reduction of GO. Reprinted with permission from ref. 21. Copyright 2012 Wiley-VCH Verlag GmbH & Co.

Nanoscale morphological characterization (via Scanning Tunnelling Microscopy and Transmission Electron Microscopy) reveals that the gradual chemical reduction of GO preferably proceeds through the removal of oxygen atoms positioned far from  $\pi$ -conjugated domains. This trend favours the formation of new, small and isolated  $sp^2$  islands, and leaves pre-existing  $sp^2$  clusters essentially unaltered (Fig. 2 d,e). As the reduction treatment proceeds, the  $I_{P2}$  Gaussian band is systematically enhanced, suggesting that it stems from the  $sp^2$  carbon. Notably,  $I_{P2}$  shifts to lower wavelengths, consistent with quantum confinement effects expected for the numerous small  $sp^2$  domains. At the same time, the initially predominant  $I_{P1}$  component monotonously decreases during deoxygenation, an effect consistent with the gradual elimination of the defect states.

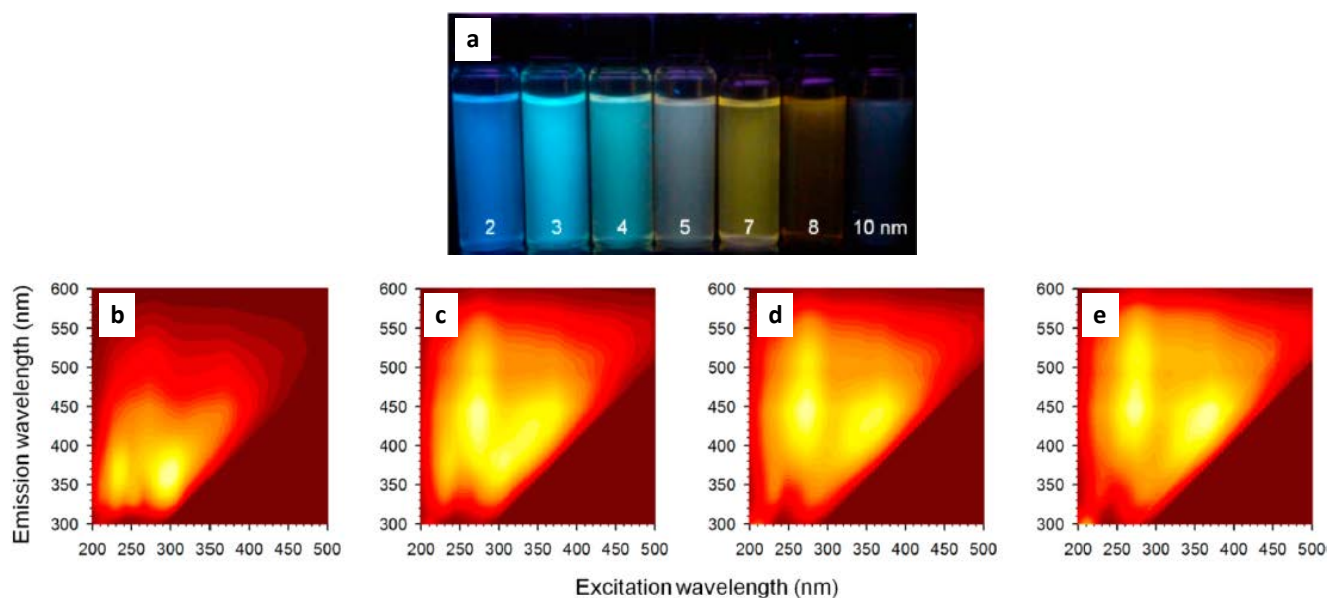
Early studies indicated that the optical properties in carbon-based materials containing different configurations of  $sp^2$  and  $sp^3$  sites are determined by the  $\pi$ - $\pi^*$  electronic levels and the band gap varies inversely with the  $sp^2$  cluster size<sup>22</sup>. Recent theoretical studies indicated that the HOMO-LUMO band gap in GQDs can be tuned from 7 eV for a single benzene ring to 2 eV for 20  $\pi$ -conjugated aromatic rings<sup>[23]</sup>, in analogy to semiconductor quantum dots.



**Figure 3.** Images of aqueous dispersions of different sized GQDs in water under (a) white and (b) UV illumination (b). (c) Dependence of the band gap as a function of the GQD size. Reprinted with permission from ref 24. Copyright 2010 Wiley-VCH Verlag GmbH & Co.

To that end, alkali assisted electrooxidation of graphite leads to a mixture of different sized GQDs that can be separated to relatively uniform particle fractions by column chromatography<sup>[24]</sup>. The aqueous dispersions of those fractions under white and UV light are shown in Fig. 3a and 3b, respectively, providing direct evidence on their quantum nature. Figure 3c shows that the band gap of GQDs decreases as the GQD size increases. In addition, hydrogen plasma treatment had no detectable effect on their PL spectra, confirming that the optical properties are attributed to quantum confinement and not to the presence of surface oxygen.

In another study, a series of well-defined GQDs with sizes from 2 to 10 nm was produced via amidative cutting of pre-oxidized graphite flakes<sup>[25]</sup>. Interestingly, their aqueous dispersions under UV light display a variety of colors from blue to brown, in a manner that critically depends on their size (Fig. 4a). The excitation matrices (a set of excitation spectra, each one collected at a fixed emission wavelength) of the GQDs with sizes 2, 4, 7, and 10 nm are shown in Figures 4b-e. The two peaks observed in each data set were attributed to the electronic transition from the ground state to the lowest and the second lowest excited state, respectively; both peaks systematically red shift upon increasing the particle size, consistent with quantum confinement effects.

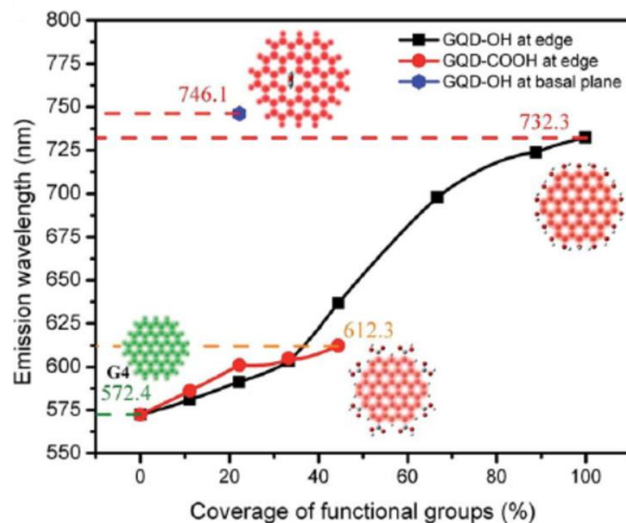


**Figure 4.** (a) Images of aqueous dispersions of different sized GQDs in water under UV illumination. The captions denote the size of GQDs in nm. (b-e) Emission matrices of aqueous dispersion of GQDs with sizes 2, 4, 7 and 10 nm, respectively. Reprinted with permission from ref 25. Copyright 2014 American Chemical Society.

There is a growing body of evidence to suggest that PL in GQDs cannot be explained in terms of quantum confinement alone and that pronounced contributions arise from edge effects, the presence of functional groups, heteroatom doping and the fraction of  $sp^2$  domains. For example, the emission wavelength of GQDs continuously shifts to higher values as a function of the -OH and -COOH coverage to edge carbon atoms (Fig. 5). In particular, for cis-coronene (G4)  $E_m=572.4$  nm, but approaches 723.3 nm for 100% edge coverage with -OH<sup>[26\*]</sup>. When two -OH groups are conjugated on the basal plane of G4, a substantial red-shift is observed, resulting in  $E_m=746.1$  nm. Experimental data confirm that green GQDs, when subjected to mild chemical reduction, become blue nanoemitters<sup>[27]</sup>, in agreement with the trends shown in Fig. 5.

Graphitic N-doping on the green G4 substantially lowers the band gap to the extent that the particles become non emissive<sup>[26\*]</sup>. Pyridinic N-doping causes a blue shift, so that 12.5% N/C ratio shifts the emission from 572.4 to 550.3 nm. Introduction of pyrrolic N atoms (N/C ratio =9.09%) at the edged five-membered rings of GQDs causes blue shift from 624.4 to 575.5 nm<sup>[26\*]</sup>.

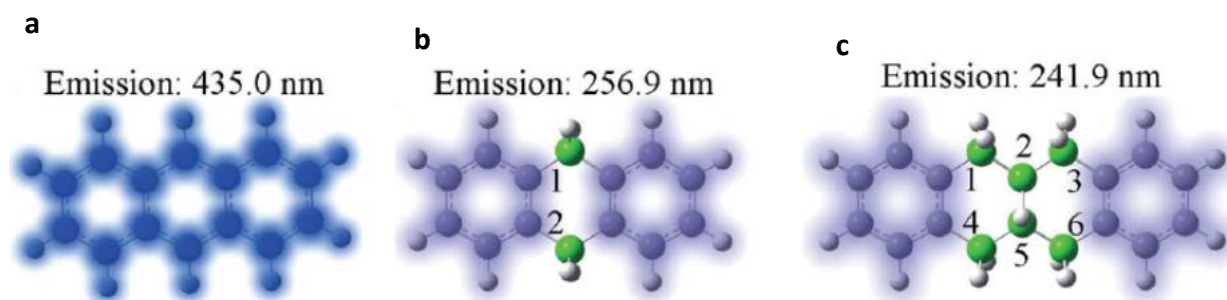




**Figure 5.** Emission wavelength of oxidized cis-coronene (G4) as a function of the coverage of –OH and –COOH groups. Reprinted with permission from ref 26. Copyright 2014, Royal Society of Chemistry.

#### 4. The common carbogenic origin

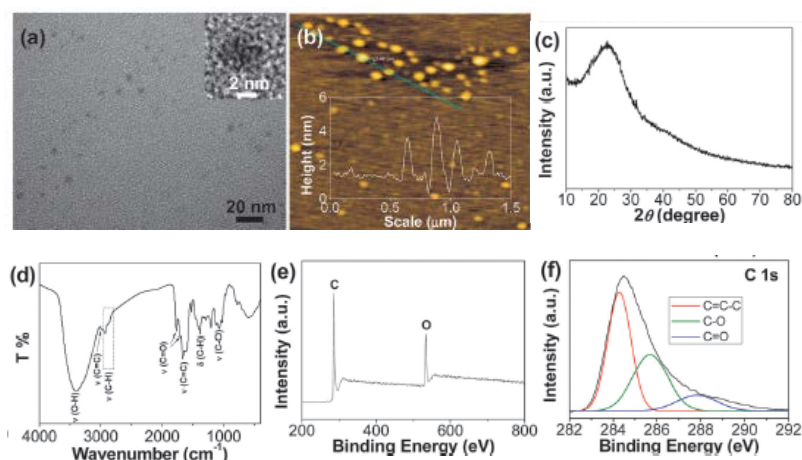
Carbogenic nanoparticles, commonly described as C-dots, are spherical nanoemitters consisting of a partially or predominantly amorphous core. In carbon materials that combine carbons with different hybridizations, it should be noted that when  $sp^3$  carbons are used to bridge two benzene rings, the emission shifts to lower wavelengths. For example, anthracene is composed of two benzene rings connected via two  $sp^2$  carbons and emits at 435 nm (Fig. 6 a), however when two benzene rings are connected via two or six  $sp^3$  carbons they emit at 256.9 and 241.9 nm, respectively<sup>[26•]</sup> (Fig. 6 b and c).



**Figure 6.** Two benzene rings joined via (a) two  $sp^2$  carbons, (b) two  $sp^3$  carbons and (c) six  $sp^3$  carbons. Reprinted with permission from ref 26. Copyright 2014, Royal Society of Chemistry.

C-dots are typically derived via hydrothermal or pyrolytic treatment of carbon-rich precursors including renewable materials such as biomass<sup>[28]</sup>, fruit juices<sup>[29]</sup> and carbohydrates<sup>[30]</sup>. The TEM images (Fig. 7a) of the polysaccharide-derived C-dots do not show any discernible lattice fringes, indicating the amorphous nature of the core<sup>[31]</sup>. TEM and AFM (Fig. 7b) both suggest particle diameters between 2-5 nm. The XRD pattern shows only a broad peak at  $2\theta=23.1^\circ$  (Fig. 7c), consistent with the amorphous carbon phase. The FTIR spectrum of C-dots indicates the presence of surface C–H, C=C, C=O, C–O and –OH groups that impart solubility in various solvents. XPS (Fig. 7e) reveals the presence of C and O, while the XPS C1s spectrum in Fig. 7f shows peaks at 284.3, 285.7 and 288.0 eV, corresponding to C=C–C, C–O and C=O groups, respectively.

**Figure 7.** C-dots hydrothermally derived from a polysaccharide: (a) TEM, (b) AFM image (inset displays the height profile along the line), (c) XRD, (d) FTIR, (e) XPS spectra (e) and XPS C1s (f) spectra of C-Dots. Reprinted with



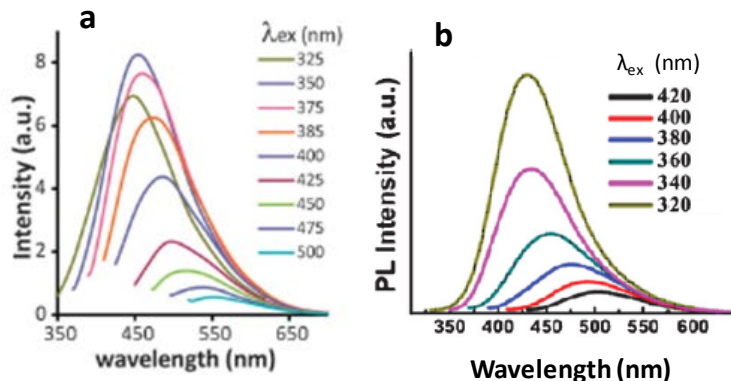
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Despite their pronounced structural differences, C-dots<sup>[32]</sup> and GQDs<sup>[14]</sup> share common PL patterns and, as the excitation wavelength increases, the emission peak is displaced to longer wavelengths and a less intense signal is recorded (Fig. 8).

It has been pointed out that defects in graphene sheets related to the presence of  $sp^3$  carbons are structurally no different from defects on the surface of C-dots and this observation explains the similarities between the PL characteristics in C-dots and GQDs<sup>[33]</sup>. We note that for GQDs, the defect-derived PL contributions can be substantially stronger compared to emissions stemming from quantized  $sp^2$  islands<sup>[33]</sup>. A recent study suggests that the green PL centers encountered in

both C-dots and GQDs originate from emissive edge states due to the combination of carboxyl groups and carbonyl groups<sup>[34]</sup>.

**Figure 8.** PL emission at different excitation wavelengths of aqueous dispersions containing: (a) C-dots derived by



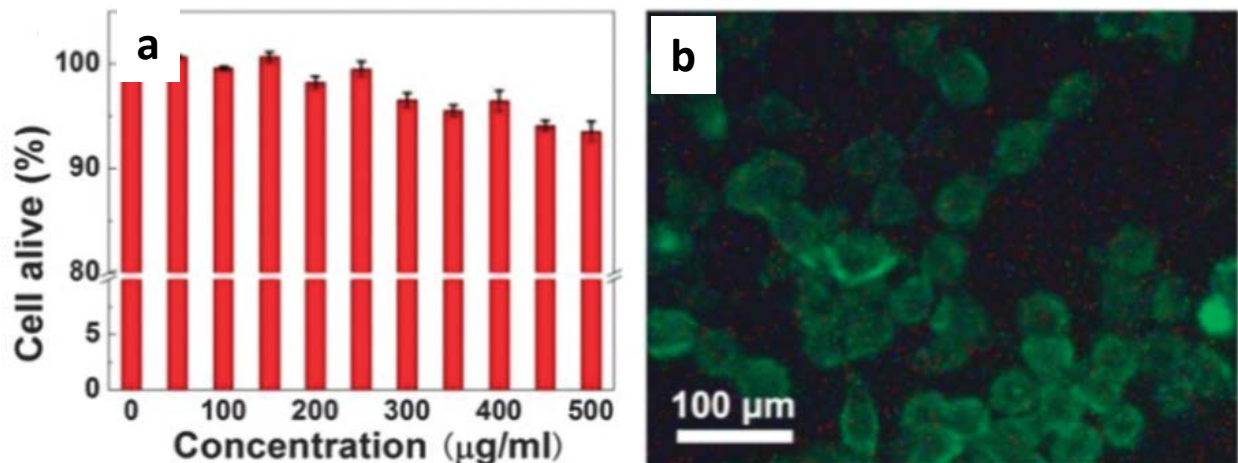
caramelization of poly(ethylene glycol) and (b) GQDs derived hydrothermally from graphene sheets. Reprinted with permission from ref 32 (Copyright 2011, Royal Society of Chemistry) and 14 (Copyright 2010 Wiley-VCH Verlag GmbH & Co), respectively.

It has been proposed that the PL behavior in C-dots arises from the radiative recombinations of surface confined photogenerated electron and holes pairs; the role of passivation agents is to stabilize the surface sites and improve the PL intensity<sup>[35]</sup>. Similar effects have been observed in GQDs<sup>[36]</sup>, further confirming the common PL origins for GQDs and C-dots.

## 5. Applications

### 5.1. Cell imaging

The viability of HeLa cells incubated for 24 h with GQDs at concentration up to 500 mg mL<sup>-1</sup> exceeds 95%<sup>[37]</sup> (Fig. 9a), suggesting that GQDs exhibit minimal cytotoxicity. Moreover, the cells display bright green PL, indicating a high degree of GQDs uptake (Fig. 9b). By virtue of their small hydrodynamic ratio and their low nonspecific protein adsorption, GQDs are promising candidates for bioimaging. Following intravenous administration GQDs remained highly fluorescent *in vivo* and followed the urine excretion<sup>[38]</sup>. GQDs that emit in the near IR have been recently reported<sup>[39]</sup> and carry the promise to circumvent issues related to cell autofluorescence and undesired photo-damage of sensitive tissue.

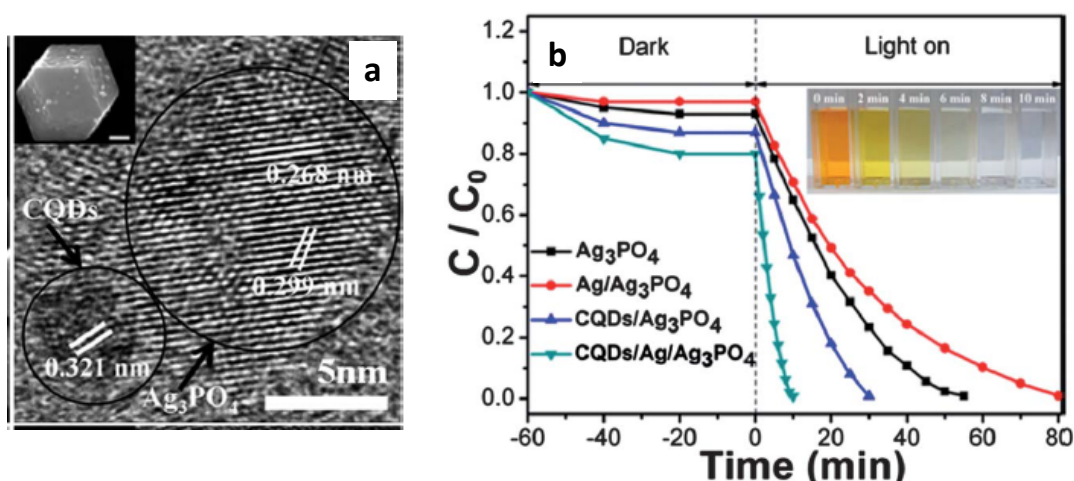


**Figure 9.** (a) Viability of HeLa cells incubated for 24h with GQDs. (b) Confocal fluorescence microscope PL image of HeLa cells incubated with 100 µg/mL GQDs ( $\lambda_{\text{ex}} = 350$  nm). Reprinted with permission from ref 37. Copyright 2015, Royal Society of Chemistry.

## 5.2. Catalysis and photocatalysis

The slow kinetics associated with the oxygen reduction reaction (ORR) impose barriers to the commercialization of fuel cells, given that the commonly used carbon supported Pt catalysts are expensive and prone to methanol poisoning. Recent work<sup>[40]</sup> indicates that the catalytic activity of N-doped GQDs is comparable to that of carbon supported Pt, with the added advantage that the former materials exhibit remarkable tolerance against methanol crossover, a situation commonly encountered in fuel cells. Boron and nitrogen co-doped GQDs deposited on graphene also exhibit improved ORR catalytic activity, showing 15 mV more positive onset potential and similar levels of current density compared to standard catalysts<sup>[41]</sup>.

As shown in Fig. 10, GQDs based composite materials (combined with  $\text{TiO}_2$ ,  $\text{Fe}_2\text{O}_3$ ,  $\text{Ag}_3\text{PO}_4$ ) exhibit significant photocatalytic activity for the decomposition of dyes, organic compounds and toxic gasses<sup>[24,42,43]</sup>. The effect reflects the synergistic action of the electron donating properties of GQDs, their upconverted luminescence and the extensive  $\pi$ - $\pi$  stacking between the aromatic rings of the organic molecules and the catalysts.



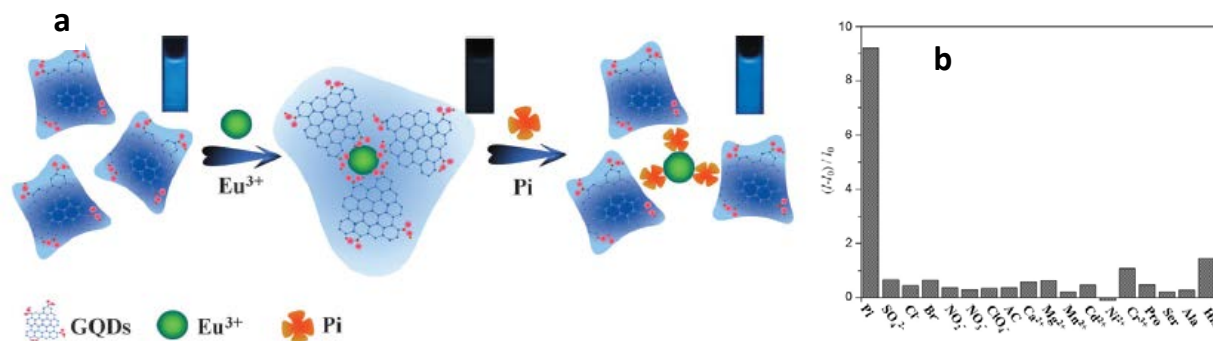
**Figure 10.** (a) TEM Images of GQDs/ Ag<sub>3</sub>PO<sub>4</sub> complex photocatalyst. (b) Decomposition rate of methyl blue (MB) in the presence of Ag<sub>3</sub>PO<sub>4</sub>, Ag/Ag<sub>3</sub>PO<sub>4</sub>, GQDs/ Ag<sub>3</sub>PO<sub>4</sub> and GQDs/Ag/Ag<sub>3</sub>PO<sub>4</sub> under visible-light. The inset shows the color changes of the MB solutions corresponding to the degradation times of 0, 2, 4, 6, 8, 10 min over GQDs/Ag/Ag<sub>3</sub>PO<sub>4</sub>. Reprinted with permission from ref 43. Copyright 2012, Royal Society of Chemistry.

### 5.3 Sensing applications

The PL properties of GQDs often exhibit selective quenching in the presence of specific ions or molecules, providing a platform for the development of high performance sensors<sup>[44]</sup>. In water, 2,4,6-trinitrotoluene (TNT) interacts via  $\pi$ - $\pi$  stacking with GQDs in a manner that suppresses PL emission via fluorescence resonance energy transfer (FRET)<sup>[45]</sup>. This system allows ultrasensitive detection of the explosive molecules with a detection limit down to 0.495 ppm TNT using only 1 mL of GQDs solution.

In another application, Eu<sup>3+</sup> ions are able to coordinate the carboxylate ions on the surface of GQDs leading to PL quenching through energy-transfer or electron-transfer processes. Because phosphate ions (Pi) display strong affinity to Eu<sup>3+</sup> they break down the GQDs-Eu<sup>3+</sup> complexes (Fig. 11a) resulting in a substantial enhancement of PL intensity<sup>[46]</sup>. The sensor is highly selective to Pi ions (Fig. 11b).

**Figure 11.** (a) Phosphate (Pi) detection based on the antagonistic interaction of  $\text{Eu}^{3+}$  with GQDs or Pi molecules. (b)



Selectivity of the PL platform for Pi detection compared to other substances. Reprinted with permission from ref 46. Copyright 2013 Wiley-VCH Verlag GmbH & Co.

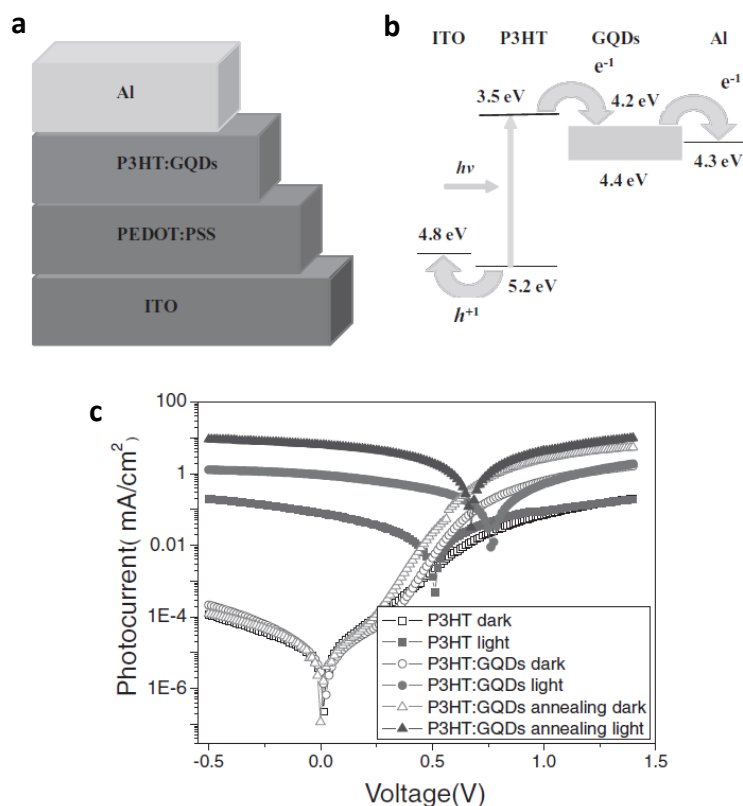
## 5.4. Energy harvesting

GQDs are ideal candidates as sensitizers in solar cells given that their absorption edge extends up to 900 nm and they exhibit one order of magnitude higher absorbance compared to standard metal complexes. In the first proof-of-concept demonstration<sup>[8]</sup>, the low current density observed pointed to the low affinity of GQDs and the  $\text{TiO}_2$ . In a recent study, N-GQDs/ $\text{TiO}_2$  hierarchical microspheres were applied in dye-sensitized solar cells, showing open circuit voltage, fill factor and power conversion efficiency 0.46V, 43% and 0.13%, respectively<sup>[47]</sup>.

In addition, GQDs have been explored as electron acceptors in order to improve the performance in bulk heterojunction (BHJ) solar cells. Incorporation of GQDs in a solar cell assembly ITO/PEDOT:PSS/P3HT:GQDs/Al [ITO, PEDOT, PSS, and P3HT stand for indium tin oxide, poly(3,4-ethylenedioxythiophene), poly(styrenesulfonate), and poly(3-hexylthiophene)] dramatically improves the power conversion efficiency (PCE) and short-circuit current ( $J_{sc}$ )<sup>[48]</sup> (Fig. 12). This behavior points to the fact that GQDs afford large p-n interfaces for charge separation.



Partially reduced GQDs derived by graphite oxide dots were found to exhibit optimal performance compared to their highly oxidized and fully reduced counterparts<sup>[49]</sup>. This behavior indicates that the overall performance reflects an interplay between enhanced light absorptivity in the highly oxidized particles that increases  $J_{sc}$  and improved conductivity in the fully reduced particles that increases the fill factors (FF). Overall, incorporation of the partially reduced GQDs resulted in PCE of 7.6% compared to 6.7% for the GQD-free device.



**Figure 12.** Schematic (a) and energy band (b) diagrams of the ITO/PEDOT:PSS/P3HT:GQDs/ Al device. c)  $J - V$  characteristic curves for the ITO/PEDOT:PSS/P3HT/Al, ITO/PEDOT:PSS/ P3HT:GQDs/Al and ITO/PEDOT:PSS/P3HT: GQDs/Al devices after annealing at 140 ° C for 10 min. Reprinted with permission from ref 48. Copyright 2011 John Wiley and Sons.

## 6. Conclusions

Historically, the rise of GQDs coincides with the rise of graphene and the two newcomers in the carbon family are expected to act synergistically in sophisticated devices. It is, nevertheless, clear

that structural defects and quantum confinement are essential for the performance of GQDs, but are detrimental for the electronic conduction of graphene.

GQDs are naturally occurring nanoparticles that can be released by coal, carbon fibers, graphite and other carbon sources. It is perhaps surprising that GQDs have been discovered only recently, two decades after the development of conventional quantum dots. Due to their strong optical absorption and excitation wavelength dependent PL emission that is not prone to photobleaching, GQDs can match and outperform highly engineered, albeit toxic, counterparts in various applications. Moreover, given that they can be produced in bulk by green methods, GQDs showcase a viable example of sustainable chemistry.

GQDs are novel nanoemitters that seem to have it all: bright fluorescence, facile and low cost preparation, minimal toxicity, supreme structural and colloidal stability. During the last years, GQDs have undoubtedly demonstrated their great potential in a series of highly demanding applications. Within the next decade, the challenge for GQDs is to open new horizons and allow new applications, currently inaccessible with existing technologies, in the fields of theranostics, catalysis, energy conversion and optical diodes.

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- Of special interest.
- Of outstanding interest.